

Protein Crystallography

Lecture 1

Introduction

Symmetry and diffraction principles

protein crystallization

Structure factor and phase problems

Experimental phasing methods

Lecture 2

Molecular replacement

Direct method

Solvent modification

Refinement and evaluation

Cross validation

Protein crystallography

Lecture 3

Applications (case study)

Use of Coot (practice in Ling lab)

Lecture 4 Short presentations by students

crystallization (seeding, new methods)

Introduction of heavy atoms

-Halides, Sel-Met, I-phe, Xe chamber etc.

MR method

MAD, SAD, MIR etc

Rfree/v alidation

Applications (structure case study)

Protein Crystallography

Crystallography

- technique reveals the three dimensional structures of molecules by X-ray diffraction.

Why X-ray

- $\lambda = 1-2 \text{ \AA}$, comparable to the C-C bond length of $\sim 1.5 \text{ \AA}$
- Interaction of X-ray with matter is through electrons which are around nuclei, so electron distribution tells us where atomic nuclei (*i.e.* atoms) are.

Why Crystal

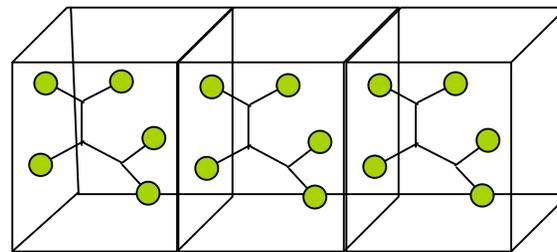
- signal from one molecule is too weak
- diffraction of hundreds of thousands molecules in a crystals can be detected

Crystal and Lattice



Crystal

3-D solid composed of an arrangement of atoms, molecules and ions that are regularly repeated throughout the volume of the solid.

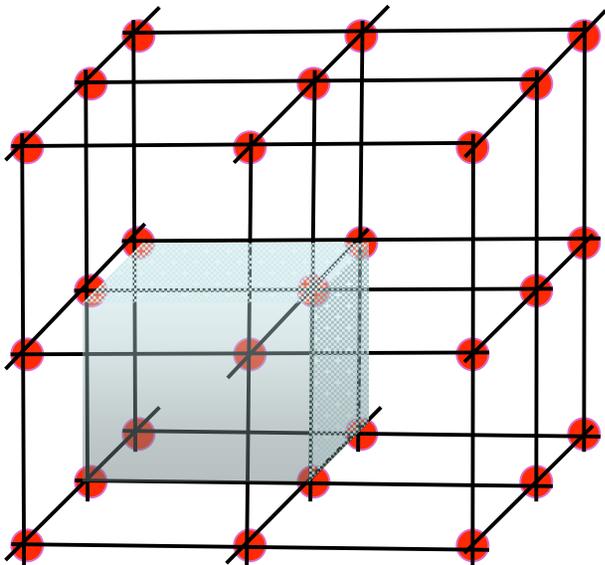


↑
Unit cell

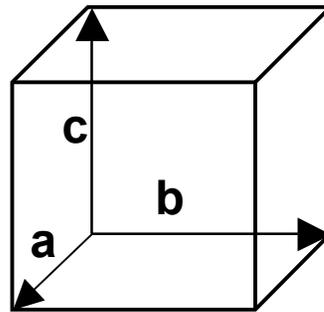
Crystal lattice: 3-D array of unit cells
periodical translational repeats of basic pattern -
“unit cell”

Crystal Lattice and Unit Cell

Three dimensional lattice

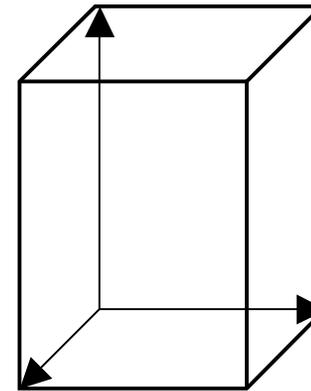


Unit cell



cubic

$a=b=c$

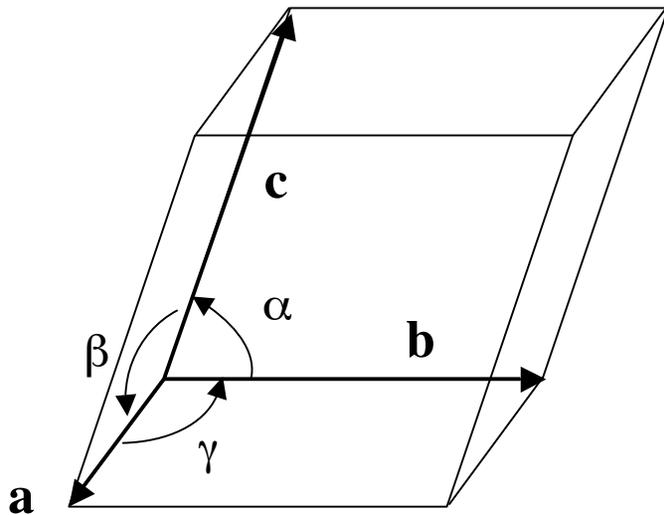


orthorhombic

$a \neq b \neq c$

Unit Cell and Symmetry

Cell defined by dimensions and angles



a, b, c, α, β, γ

7 crystal systems:

Triclinic

$$a \neq b \neq c, \alpha \neq \beta \neq \gamma \neq 90^\circ$$

Monoclinic

$$a \neq b \neq c, \beta \neq 90^\circ$$

Orthorhombic

$$a \neq b \neq c, \alpha = \beta = \gamma = 90^\circ$$

Tetragonal

$$a = b \neq c, \alpha = \beta = \gamma = 90^\circ$$

Trigonal

$$a = b \neq c, \alpha = \beta = 90^\circ, \gamma = 120^\circ$$

Hexagonal

$$a = b \neq c, \alpha = \beta = 90^\circ, \gamma = 120^\circ$$

Cubic

$$a = b = c, \alpha = \beta = \gamma = 90^\circ$$

Crystallography 101

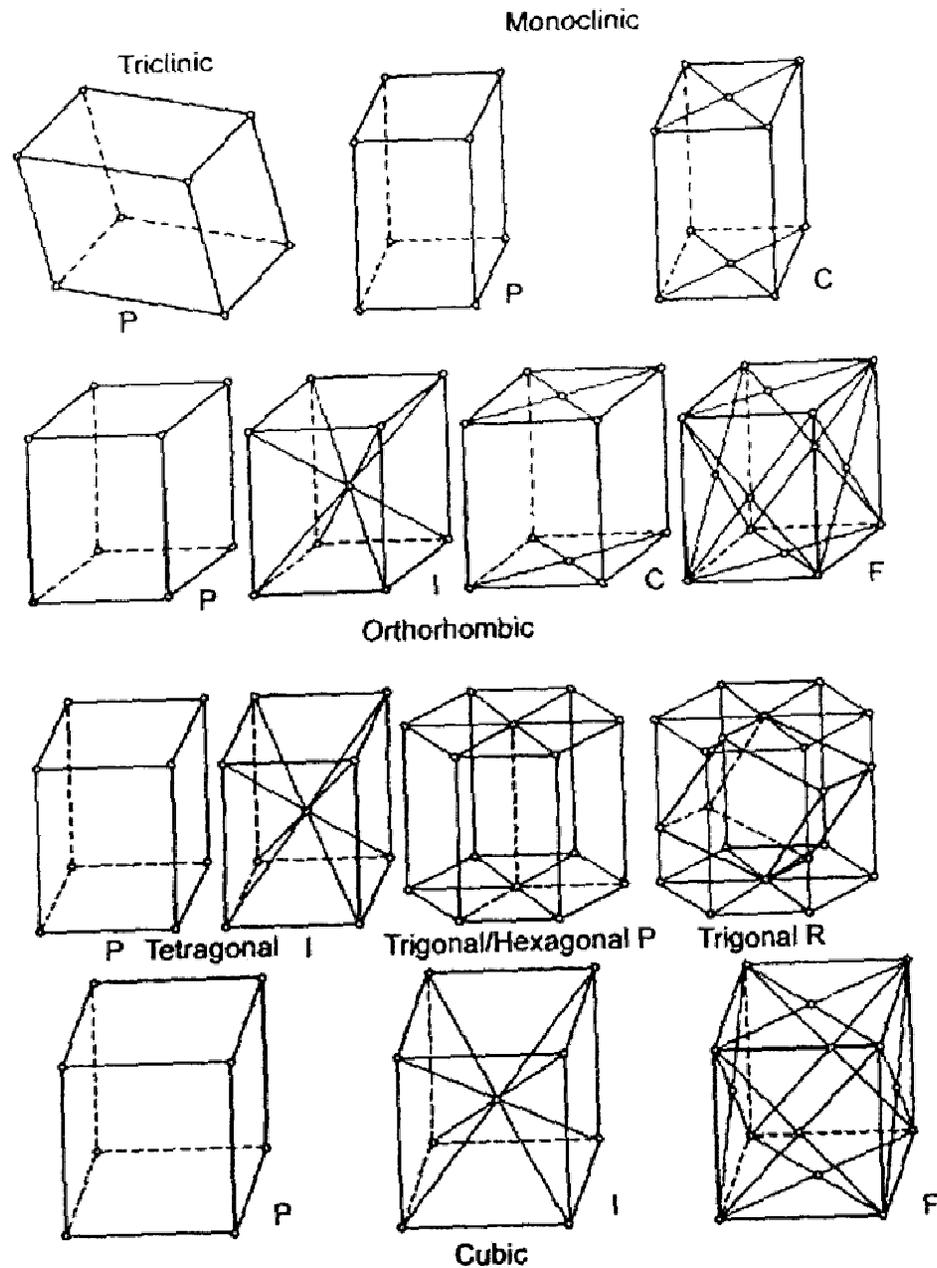
<http://ruppweb.dyndns.org/Xray/101index.html>

14 Bravis Lattices

4 lattice types
P, I, C, F

7 crystal systems

P- Primitive
I- body centered
C- C centered
F- Face centered



Symmetry and Unit Cell

- **Unit cell:** A minimum repeating unit (parallelepiped) in a crystal lattice for translational repeats in 3-D space.
 - a translational symmetry
 - cell parameters: $a, b, c, \alpha, \beta, \gamma$, in right-handed axis systems
- **Symmetry and Space group**
 - relationships of identical objects in 3-D space (operations, elements)
e.g. rotations: 2, 3, 4, 6 fold axes
mirror: m
screw axis: $2_1, 3_1, 3_2, 4_1, 4_2, 4_3, 6_1, 6_2, etc$
 - a set of symmetry operations existing in a certain crystal
~230 space groups in total
 - 65 space groups for protein crystal

Space group and International table

- **Symmetry contents**
 - International tables for crystallography (Vol. I)
- **Asymmetric unit (AU)**
 - multiplicity (Z)
 - e.g. P21212, four AUs,
so $V_{\text{au}} = 1/Z \times V_{\text{unit cell}}$

Space group P21212

$P2_12_12$

No. 18

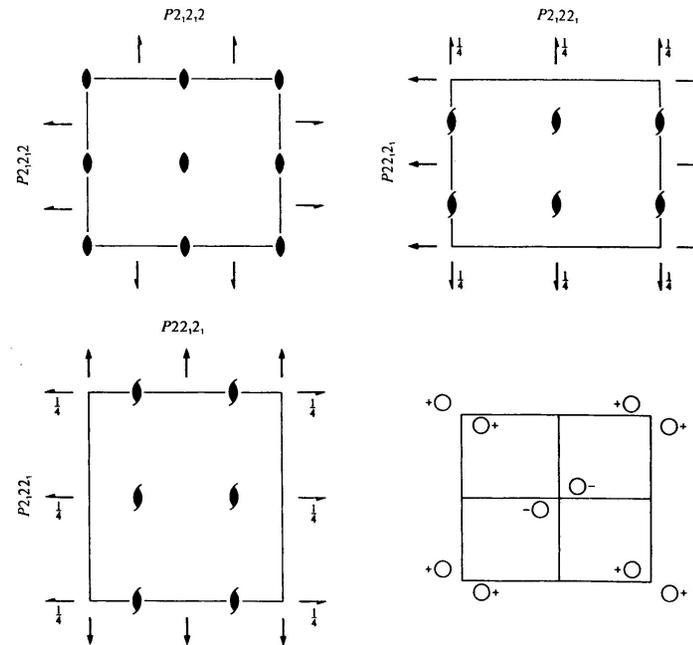
D_2^3

$P2_12_12$

222

Orthorhombic

Patterson symmetry $Pmmm$



Origin at intersection of 2 with perpendicular plane containing 2₁ axes

Asymmetric unit $0 \leq x \leq \frac{1}{2}; 0 \leq y \leq \frac{1}{2}; 0 \leq z \leq 1$

Symmetry operations

(1) 1 (2) 2 0,0,z (3) 2(0; $\frac{1}{2}$;0) $\frac{1}{2}$,y,0 (4) 2($\frac{1}{2}$,0,0) $x,\frac{1}{2}$,0

Space group P21212

CONTINUED

No. 18

$P2_12_12$

Generators selected (1); $t(1,0,0)$; $t(0,1,0)$; $t(0,0,1)$; (2); (3)

Positions

Multiplicity,
Wyckoff letter,
Site symmetry

Coordinates

Reflection conditions

4 *c* 1 (1) x, y, z (2) \bar{x}, \bar{y}, z (3) $\bar{x} + \frac{1}{2}, y + \frac{1}{2}, \bar{z}$ (4) $x + \frac{1}{2}, \bar{y} + \frac{1}{2}, \bar{z}$

General:

$h00: h = 2n$

$0k0: k = 2n$

Special: as above, plus

2 *b* .. 2 $0, \frac{1}{2}, z$ $\frac{1}{2}, 0, \bar{z}$

$hk0: h + k = 2n$

2 *a* .. 2 $0, 0, z$ $\frac{1}{2}, \frac{1}{2}, \bar{z}$

$hk0: h + k = 2n$

Symmetry of special projections

Along [001] $p2gg$

$\mathbf{a}' = \mathbf{a}$ $\mathbf{b}' = \mathbf{b}$

Origin at 0, 0, z

Along [100] $p2mg$

$\mathbf{a}' = \mathbf{b}$ $\mathbf{b}' = \mathbf{c}$

Origin at $x, \frac{1}{2}, 0$

Along [010] $p2gm$

$\mathbf{a}' = \mathbf{c}$ $\mathbf{b}' = \mathbf{a}$

Origin at $\frac{1}{2}, y, 0$

Maximal non-isomorphic subgroups

I [2] $P12_11 (P2_1, 4)$ 1; 3

[2] $P2_111 (P2_1, 4)$ 1; 4

[2] $P112 (P2, 3)$ 1; 2

IIa none

IIb [2] $P2_12_12_1 (c' = 2c)$ (19)

Maximal isomorphic subgroups of lowest index

IIc [2] $P2_12_12 (c' = 2c)$ (18); [3] $P2_12_12 (a' = 3a \text{ or } b' = 3b)$ (18)

Minimal non-isomorphic supergroups

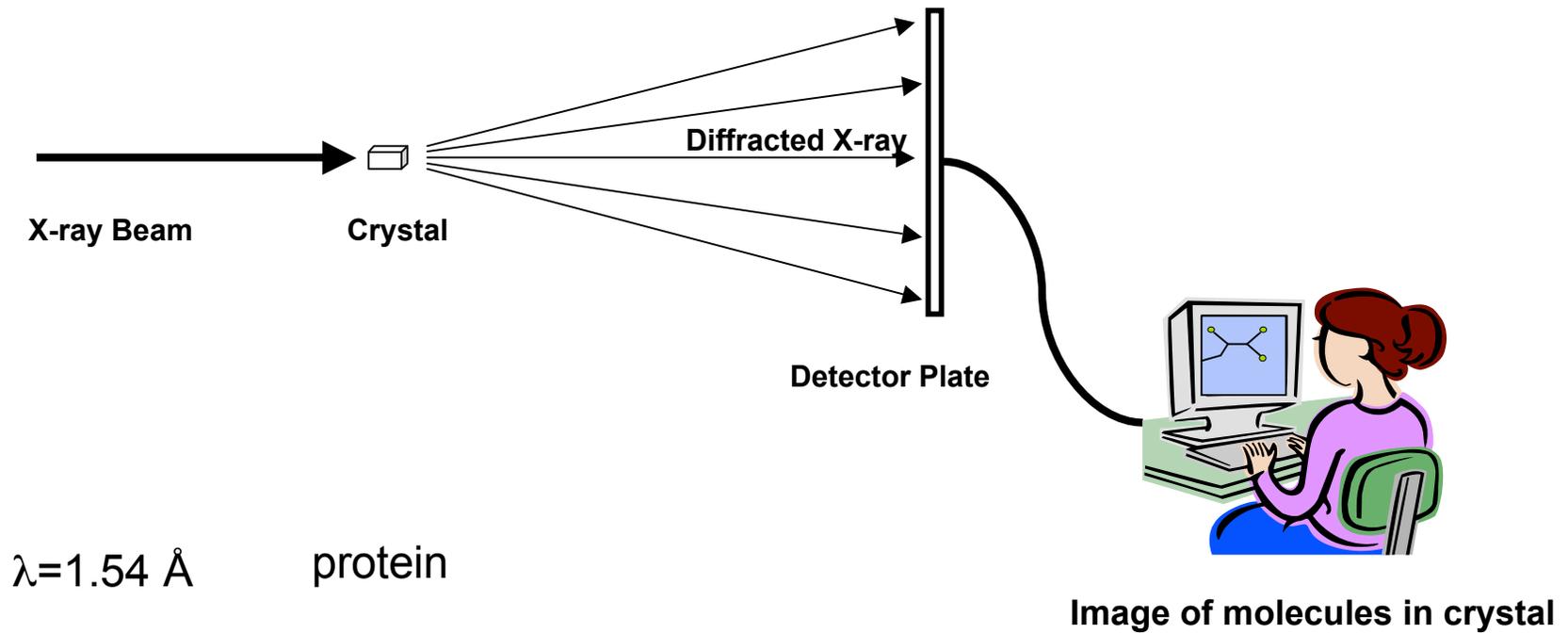
I [2] $Pbam$ (55); [2] $Pccn$ (56); [2] $Pbcm$ (57); [2] $Pnmm$ (58); [2] $Pmnm$ (59); [2] $Pbcn$ (60); [2] $P4_22_12$ (90); [2] $P4_22_12$ (94);

[2] $P4_2m$ (113); [2] $P4_2c$ (114)

II [2] $A2_122 (C222_1, 20)$; [2] $B2_12_12 (C222_1, 20)$; [2] $C222$ (21); [2] $I222$ (23); [2] $P22_12 (a' = \frac{1}{2}a)$ ($P222_1, 17$);

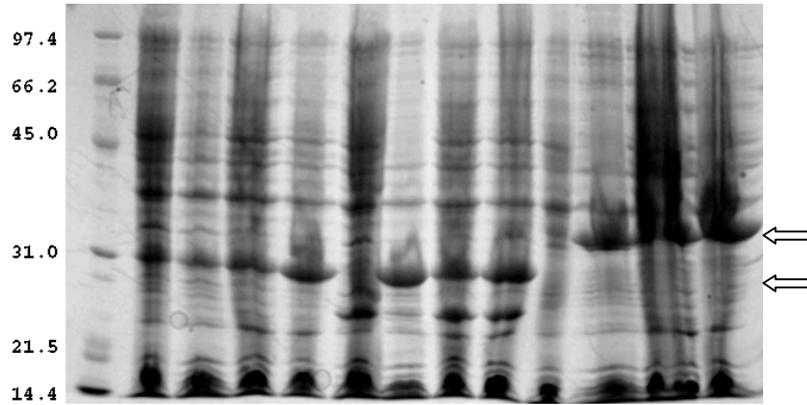
[2] $P2_122 (b' = \frac{1}{2}b)$ ($P222_1, 17$)

X-ray crystallography

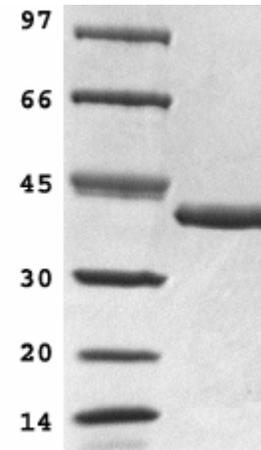


General Steps

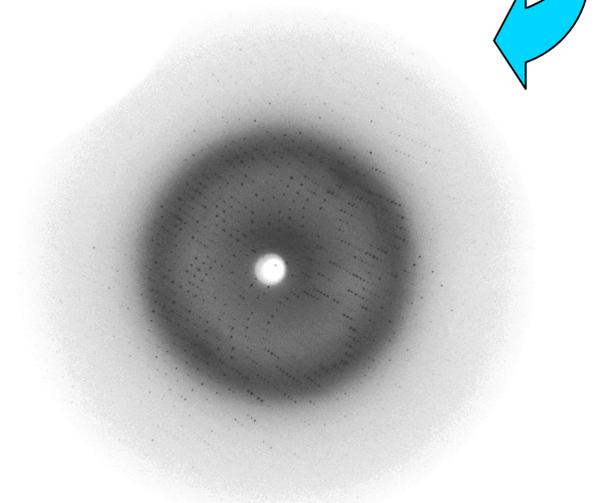
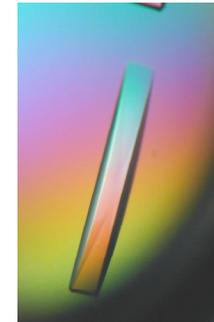
Expression



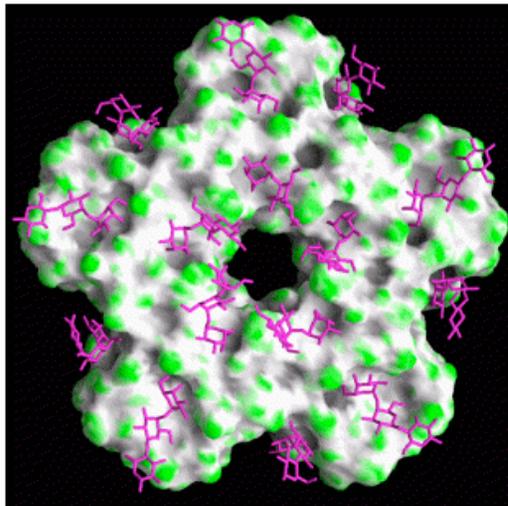
Purification



Crystallization



X-ray diffraction



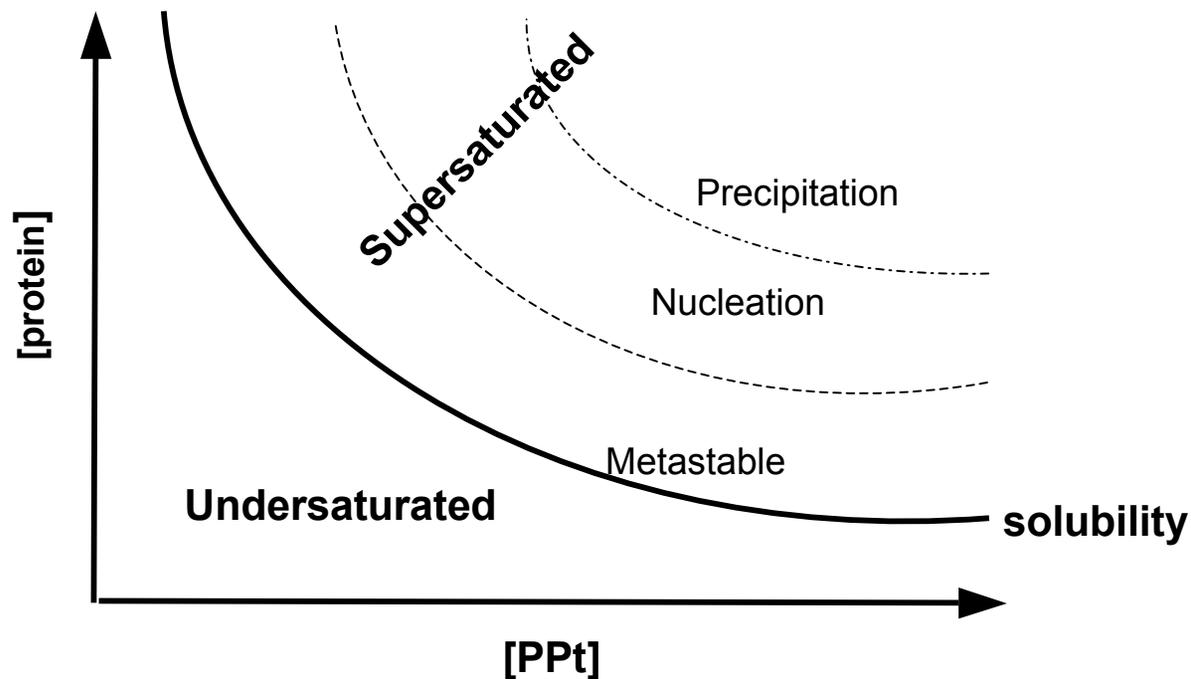
Structural analysis

Major Steps in Crystallography

- **Grow crystals**
- **Measure the intensities of the diffraction pattern**
- **Compute an electron density map**
- **Interpret the electron density map and construct model**
- **Refine molecular model to fit experimental observations**

Protein Crystallization

- Principles of protein solubility



Crystallization:

- To create supersaturation in protein solution by Increasing protein concentration beyond their saturation points gently

Crystallization

Crystallization

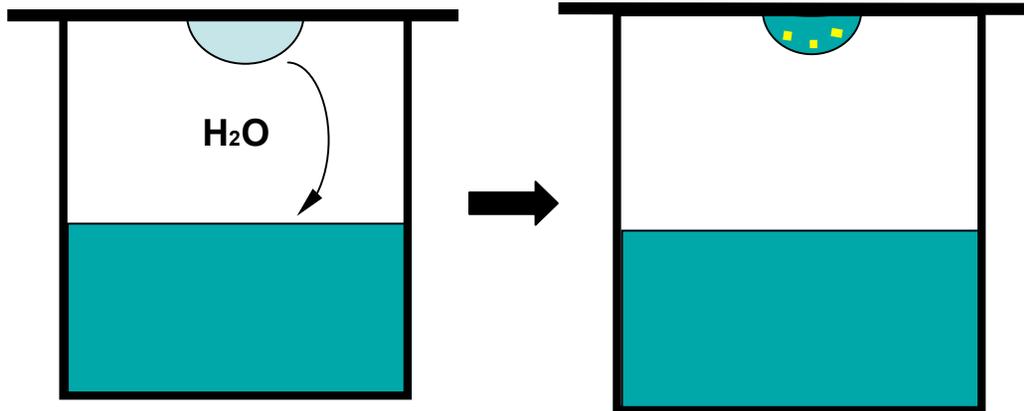
- precipitate protein from solution in organized fashion.
- by increasing protein concentration $> S_o$ (solubility)

Methods

- vapor diffusion

2ul protein +
2ul well solution

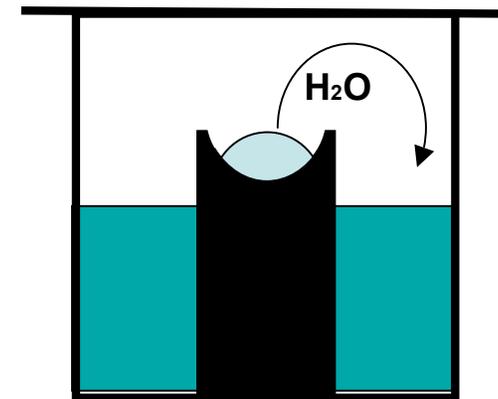
Hanging drop



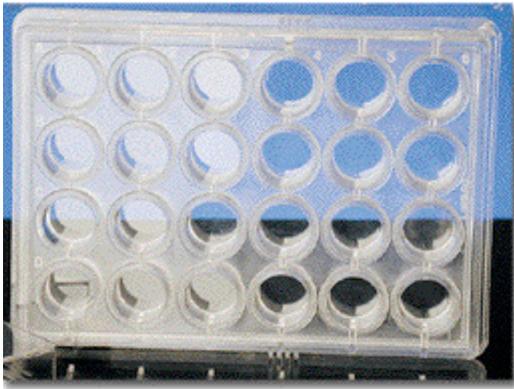
$$[ppt]_{drop} < [ppt]_{well}$$

$$[ppt]_{drop} = [ppt]_{well}$$

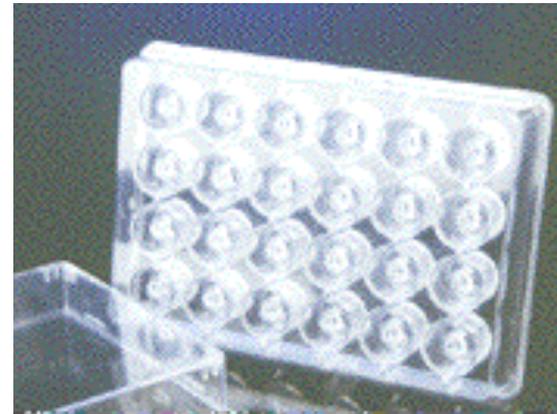
Sitting drop



Crystallization Plates



A Linbo plate for hanging drop

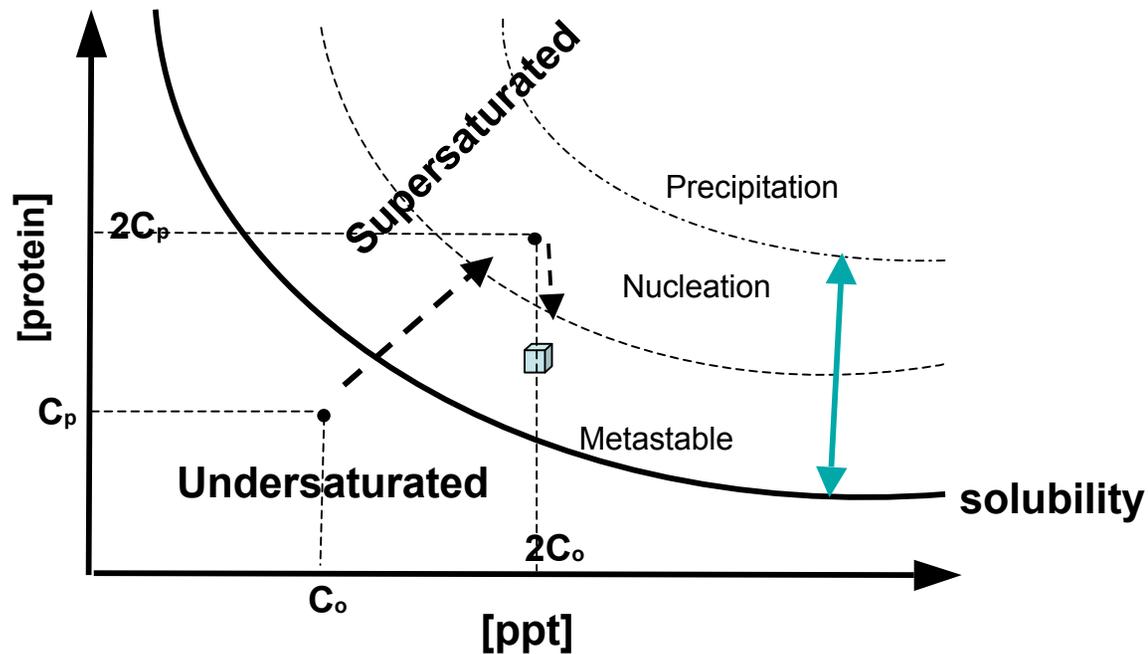


24-well plate for sitting drop

Hampton site: <http://hamptonresearch.com/support/>

Protein Crystallization

- **Screening/optimization**
crystal growth depends on:
type of precipitants, [ppt], pH, T, salts, organic solvents, etc.



Purpose: to find the conditions that nucleation and metastable regions exist in the phase diagram and hit the right concentrations.

Seeding

Problems:

- Nucleation hard to occur or to be controlled
- Supersaturation for nucleation is too high to slow crystal growth that leads to well-ordered , large-sized crystals (small crystal clusters)
- lack of reproducibility

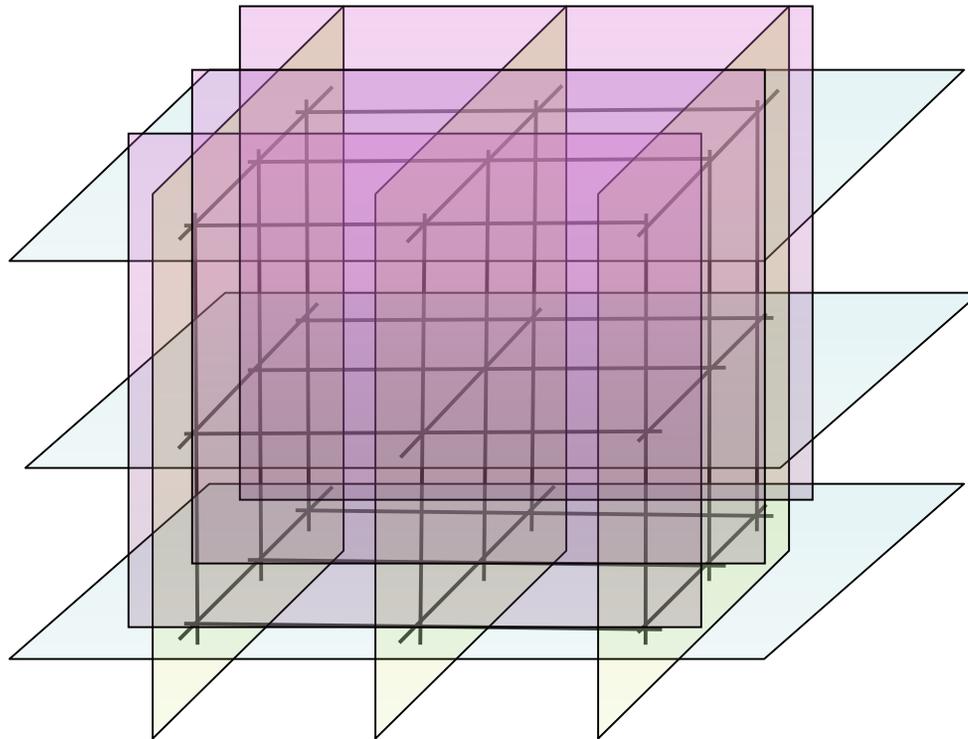
Solutions:

- Skip the nucleation stage
- Separate the nucleation and crystal growth stages
- increase size of crystals
- reduce showers of crystals
- improve crystal quality

Major Steps in Crystallography

- **Grow crystals**
- **Measure the intensities of the diffraction pattern**
- **Compute an electron density map**
- **Interpret the electron density map and construct model**
- **Refine molecular model to fit experimental observations**

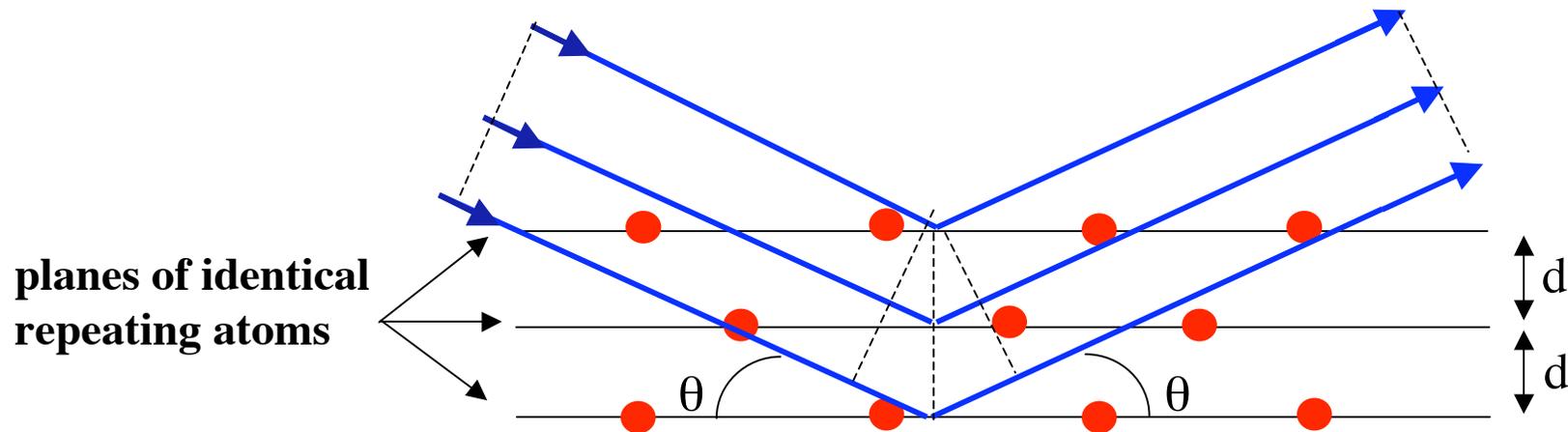
Lattice and Planes



Assembly of a set of planes

Bragg's Law

X-rays are diffracted by lattice planes of atoms when the path difference equal to integral (n) folds of the wavelength.



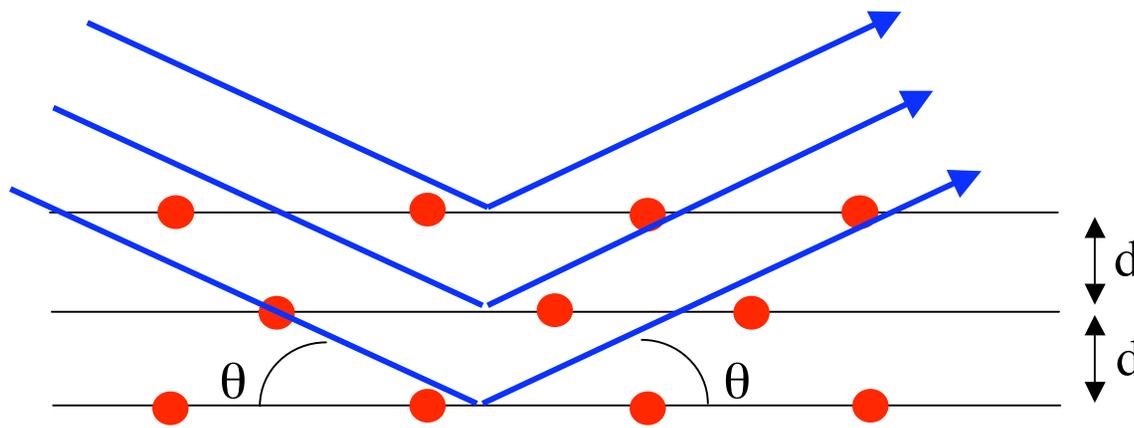
$\theta = \text{incident light } \angle = \text{diffracted light } \angle$

The diffraction is analog to reflection, so it also called X-ray reflection.

$$2d\sin\theta = n\lambda$$

Optical path difference \rightarrow Wavelength of X-ray

Bragg's Law and Data collection



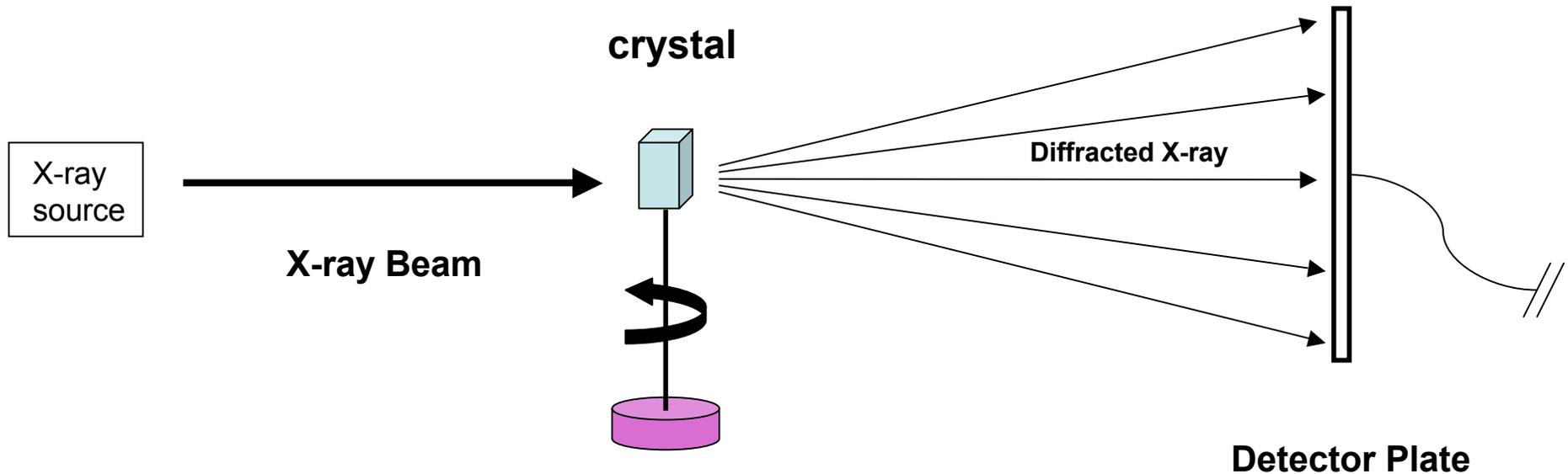
$$2d\sin\theta = n\lambda$$

- n = integer
- λ = constant
- d = distance between planes or resolution
- θ = incident/diffraction angle

- Since n is integral, the diffraction is discrete

- Diffraction occurs only when X-ray with certain incident angle θ that meets the Bragg's Law.

Diffraction Data Collection

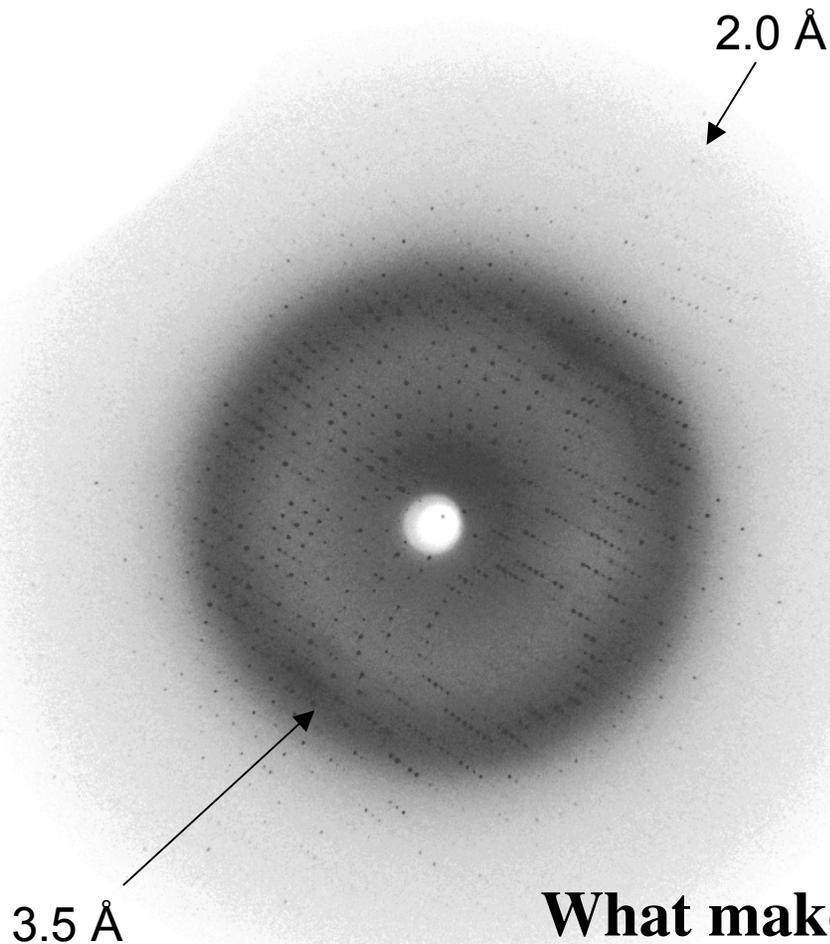


As the crystal rotates along the axis perpendicular to X-ray beam, lattice planes in different orientations will be in diffraction angles (θ) to X-ray; *i.e.*, when sets of planes meet the condition of Bragg's Law, diffraction occurs.

During the rotation of a crystal, many sets of planes will be in diffraction angles, so we collect lots reflections

X-ray diffraction

Diffraction Pattern



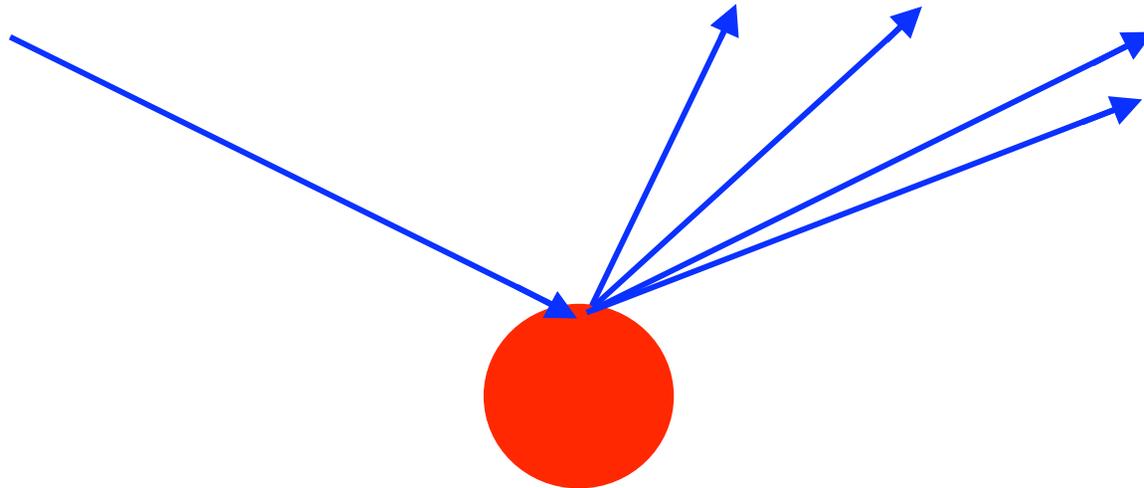
Spots are reflections:

- the positions determined by the cell parameters
- the intensities (I) determined by the protein structure
- Intensity with hkl index (I_{hkl})
- data is a set of $I_{hkl}(s)$
- resolution at the edge, θ_{\max}

What makes the intensity different in spots?

X-ray scattered by atom

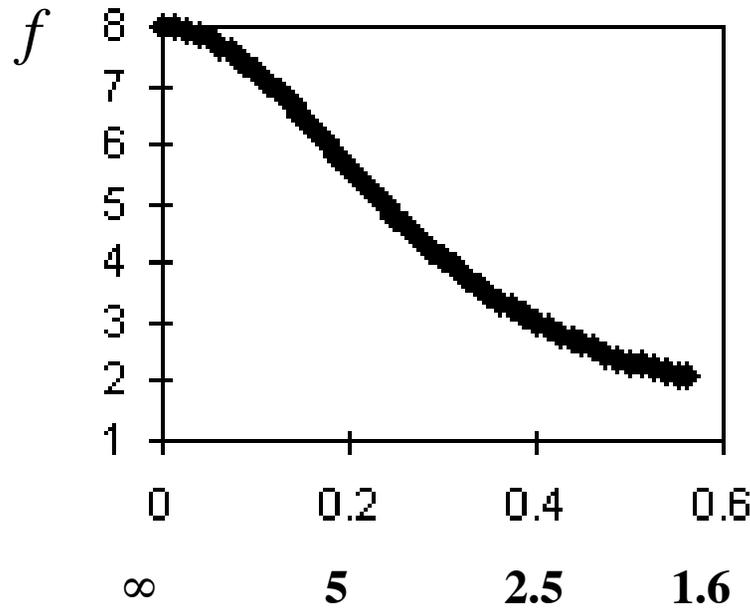
Scattering is in 3D



Diffraction is actually from atoms in planes

Intensities have 3D information

Atomic Scattering Factor



$$f^o(\sin\theta/\lambda) = \sum_{i=1}^4 a_i \cdot e^{-b_i(\sin\theta/\lambda)^2} + c$$

Oxygen atomic scattering factor
 $Z = 8$

Diffraction from different atoms - scattering factor f_j

based on atom, numbers of electrons \rightarrow At zero degree equal to Z

Larger atoms - more electrons diffract strongly

Small atoms - fewer electrons diffract weakly

Structure Factor

Atomic structure factor

$$f_{(hkl)} = f_{(j)} \exp [2\pi * i(hx_{(j)} + ky_{(j)} + lz_{(j)})]$$

hkl = index from diffraction pattern

$f_{(j)}$ = atomic scattering factor for atom j

x, y, z - coordinates of an atom

For a group of atoms (i.e. molecules in unit cells)

$$F_{(hkl)} = \sum_{j=1}^{\text{atoms}} f_{(j)} \exp [i2\pi(hx_{(j)} + ky_{(j)} + lz_{(j)})]$$

$F_{(hkl)}$ = structure factor

Structure factor represents x-ray diffraction of molecules in a unit cell and is determined by all the atoms in the unit cell and their positions

Structure Factor and Intensity

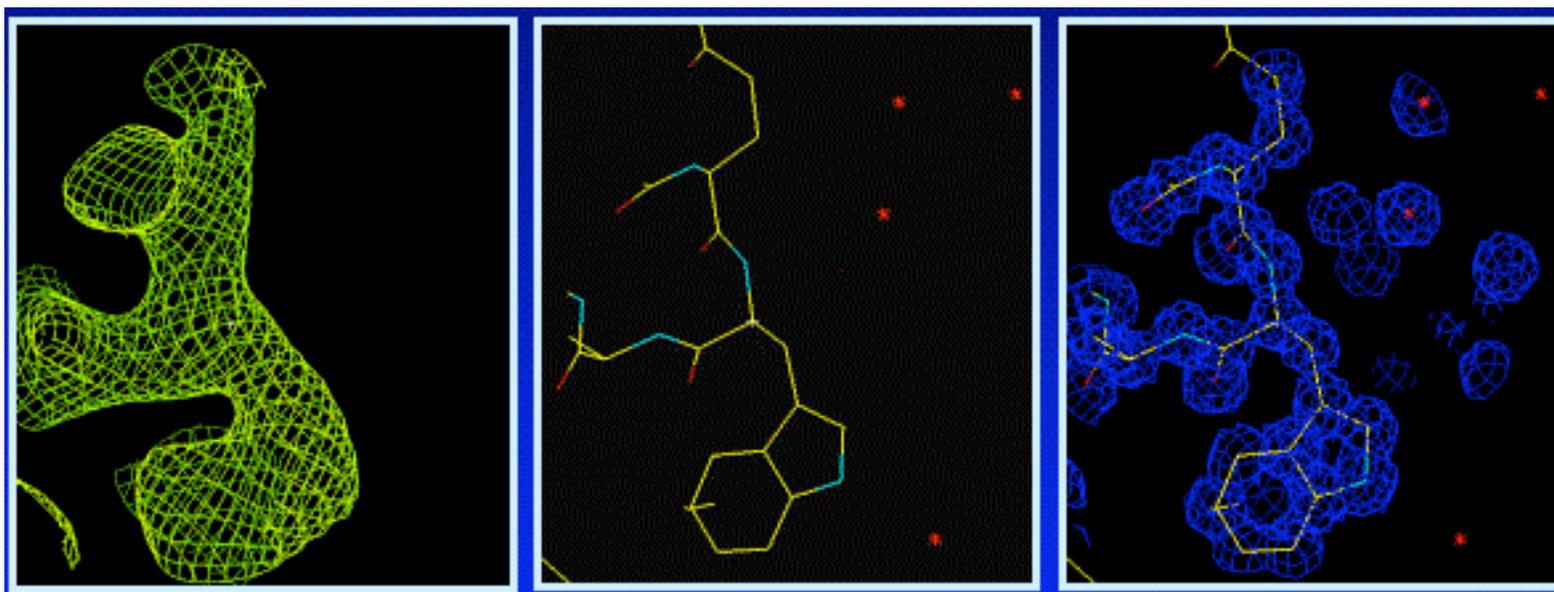
$$I_{hkl} = k |F_{hkl}|^2$$

$$|F| = k I^{1/2}$$

The amplitude of F_{hkl} is proportional to the square root of the intensity, so I_{hkl} determined by molecular structure in the crystal.

The amplitudes ($|F_{hkl}|$) can be obtained from the measured intensities (I_{hkl}) - X-ray diffraction data.

Electron Density Map



Initial map

model

refined model and map

Crystallography 101

<http://ruppweb.dyndns.org/Xray/101indexhtml>

Electron Density Calculation and Phase Problem

$$\rho(x, y, z) = \frac{1}{V} \sum_h \sum_k \sum_l F_{hkl} \exp(-2\pi i(hX + kY + lZ))$$

X, Y, Z: any point in the unit cell

$$F_{hkl} = |F_{hkl}| \exp(i \alpha_{hkl})$$

From diffraction

phase of F_{hkl} : $\alpha_{hkl} = 2\pi(hx_j + ky_j + lz_j)$

x_j, y_j, z_j : coordinates for atom j , unknown

Ways of obtaining phases (solving structure)

Guess:

- Molecular Replacement (homology > 30%)

Heavy atom methods (experimental phasing)

- MIR: multiple isomorphous replacement
- MAD: multiple wavelength anomalous diffraction
- SIRAS: single isomorphous replacement with anomalous scattering
- SAD: single wavelength anomalous diffraction

Direct method

- mathematical/statistical method

Introduction of heavy atoms into protein crystals

- Heavy metal soaking (Hg, Au, Pt, Cs... etc)**
 - traditional methods**
- Se-Met incorporation**
 - use autotrophic strains or deplete Met in media**
- Xe into crystals by Xe gas chamber**
- Halide ions soaking**
- I-Phe incorporation (modified genetic code)**

Multiple data sets for experimental phasing

Native protein crystal: $|F_p|$

Plus:

MIR: heavy atom derivative crystal 1: $|F_{PH1}|$

“ 2: $|F_{PH2}|$

.....

“ n: $|F_{PHn}|$

MAD: heavy atom derivative crystal at λ_1 : $|F_{PH1}|$

“ λ_2 : $|F_{PH2}|$

.....

“ λ_4 : $|F_{PH4}|$

MAD data sets collected at synchrotron centers

Heavy atoms positions determined by different Patterson methods

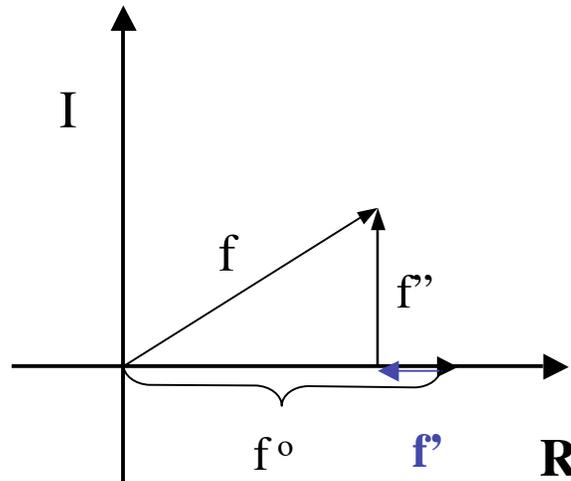
$$|F_H| = |F_{PH}| - |F_P|$$

Different Patterson function:

$$P(u,v,w) = 1/v \sum_h \sum_k \sum_l |F_H|^2 \exp [-2\pi i(hu + kv + lw)]$$

The locations of the Patterson peaks represent vectors between heavy atoms, thus, the position of the heavy atoms can be deduced from the Patterson peaks. F_H derived from heavy atom positions.

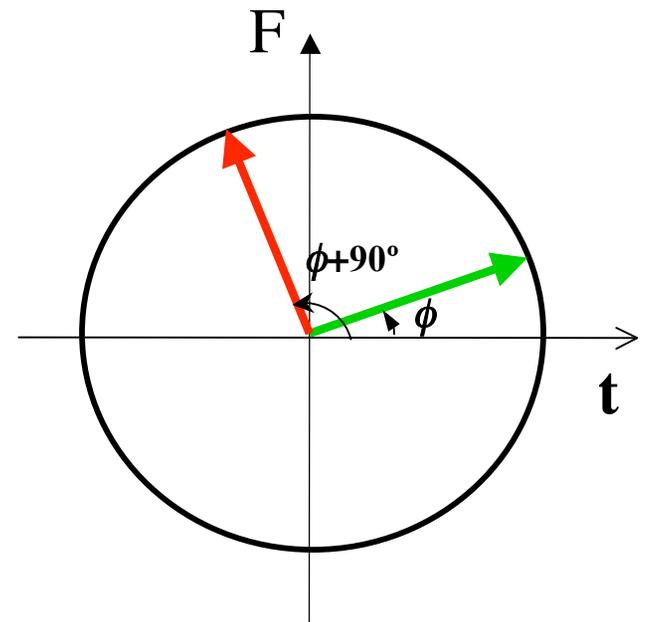
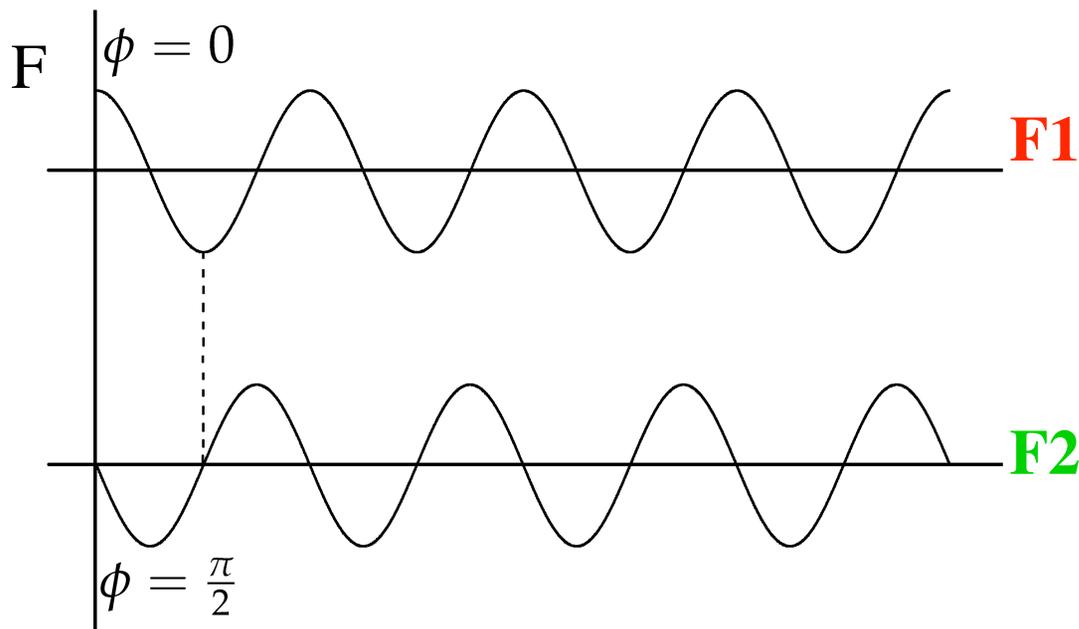
Anomalous scattering



$$\mathbf{f} = \mathbf{f}^0 + \mathbf{f}' + i\mathbf{f}''$$

f' and f'' are wave length dependent and detectable only near absorption edges
 $f = f^0$ when there is no anomalous scattering

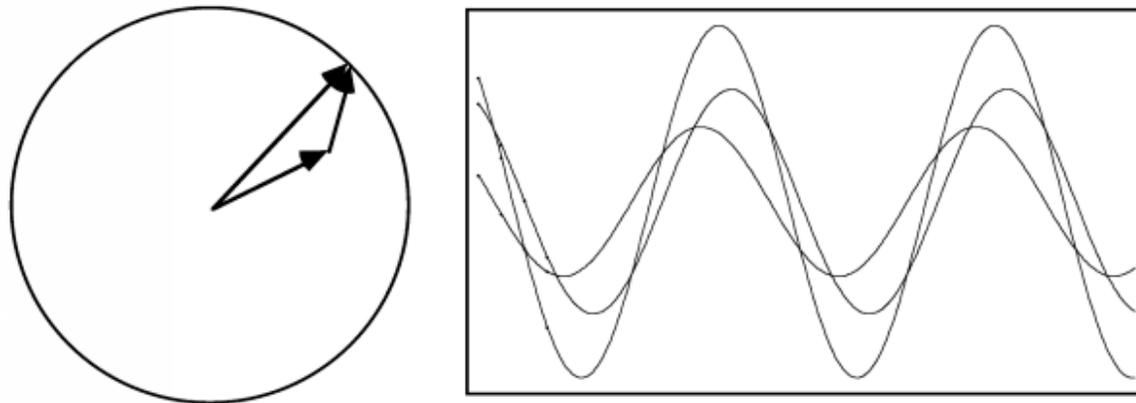
Periodic Function as a Wave



$$F = A \cos(\omega t - \phi)$$

$$\omega t = 2\pi \nu t$$

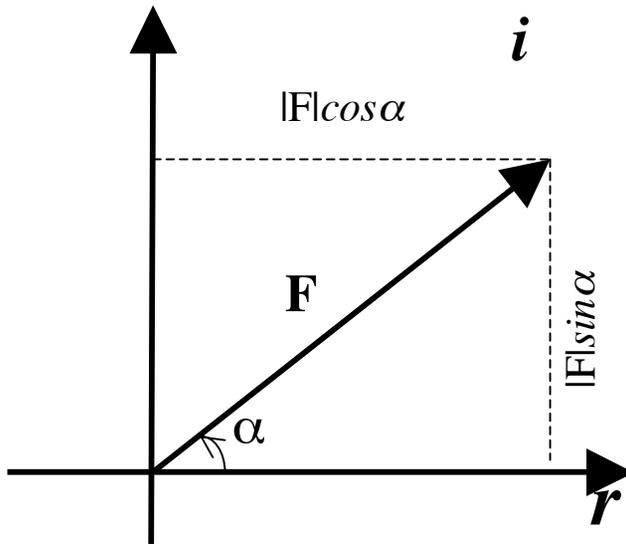
Adding waves as vectors



Cambridge X-ray course

<http://www-structmed.cimr.cam.ac.uk/course.html>

Argand diagram of the vector of a wave

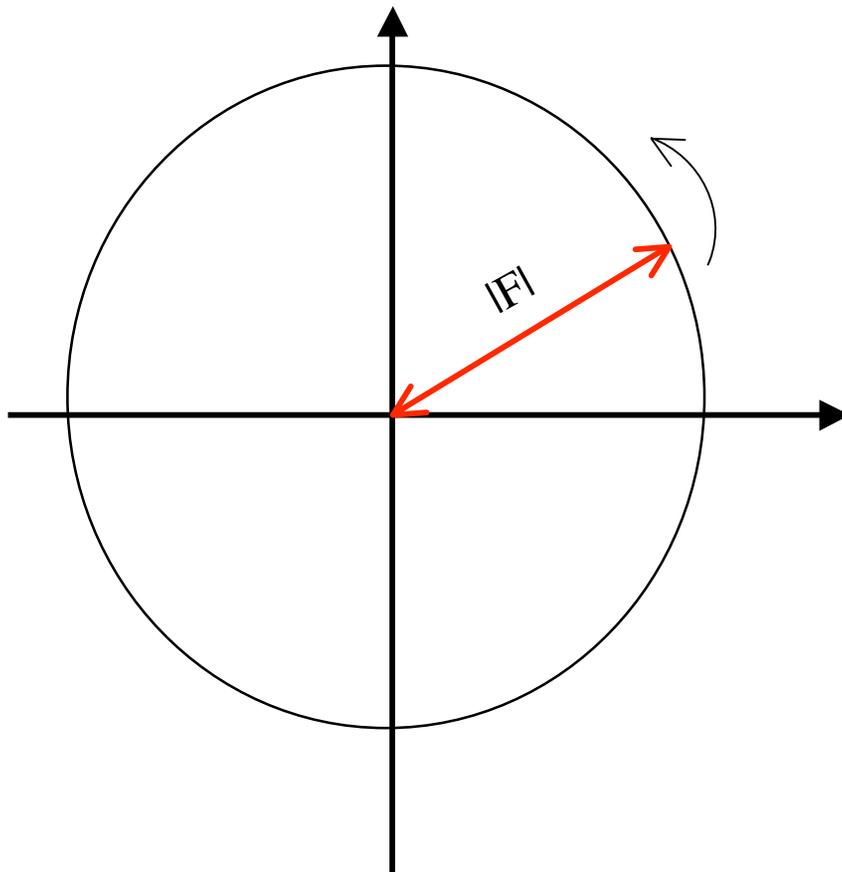


$$\mathbf{F} = |\mathbf{F}|\cos\alpha + i|\mathbf{F}|\sin\alpha$$

Euler Formula: $e^{\pm i\alpha} = \cos \alpha \pm i \sin \alpha$

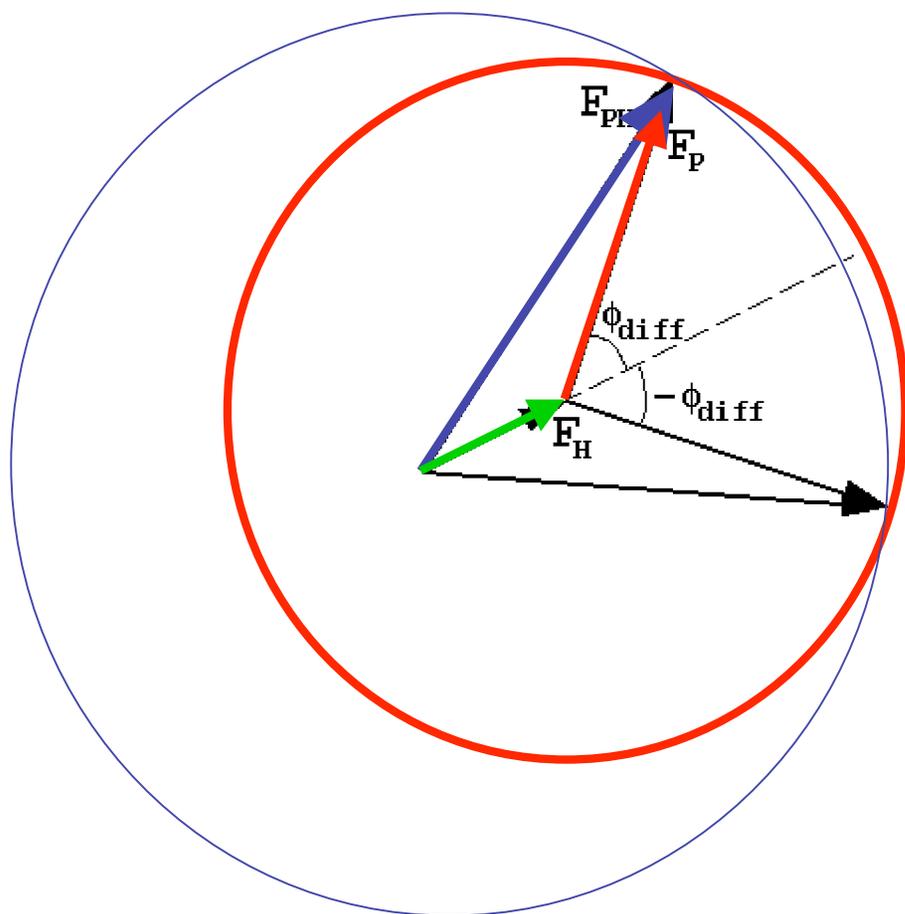
$$\mathbf{F} = |\mathbf{F}|e^{i\alpha}$$

Probability of phases of unknown structure



$$|F| = k I^{1/2}$$

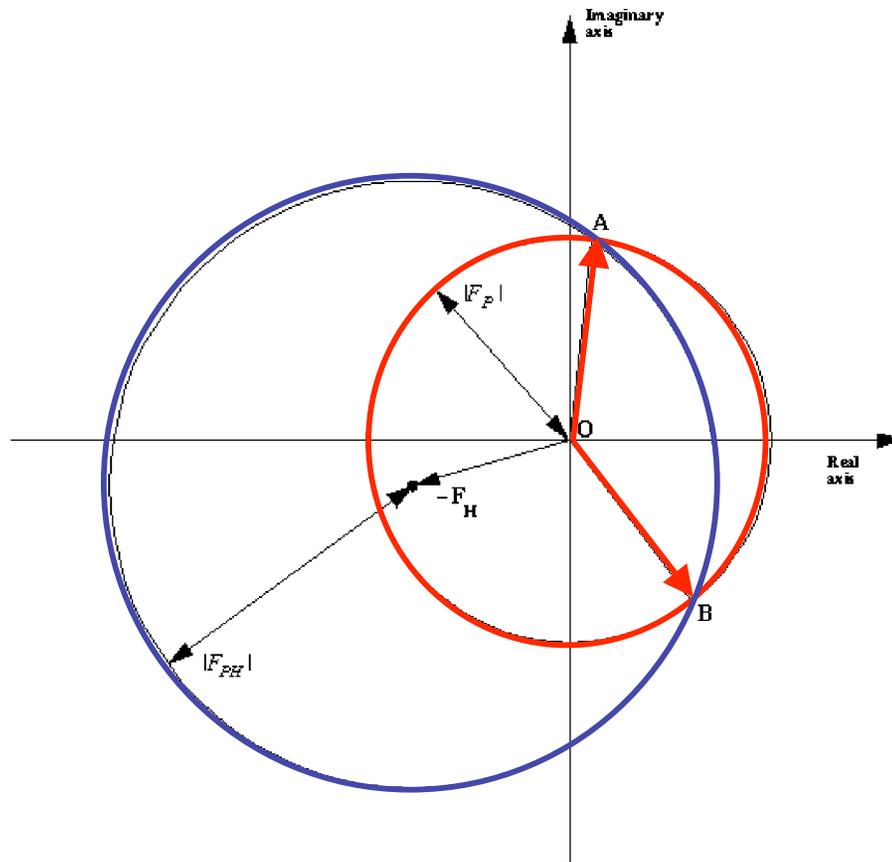
Phases of a heavy atom derivative



F_{PH}: heavy atom derivative
F_P: protein
F_H: heavy atoms

$$\mathbf{F}_{PH} = \mathbf{F}_P + \mathbf{F}_H$$

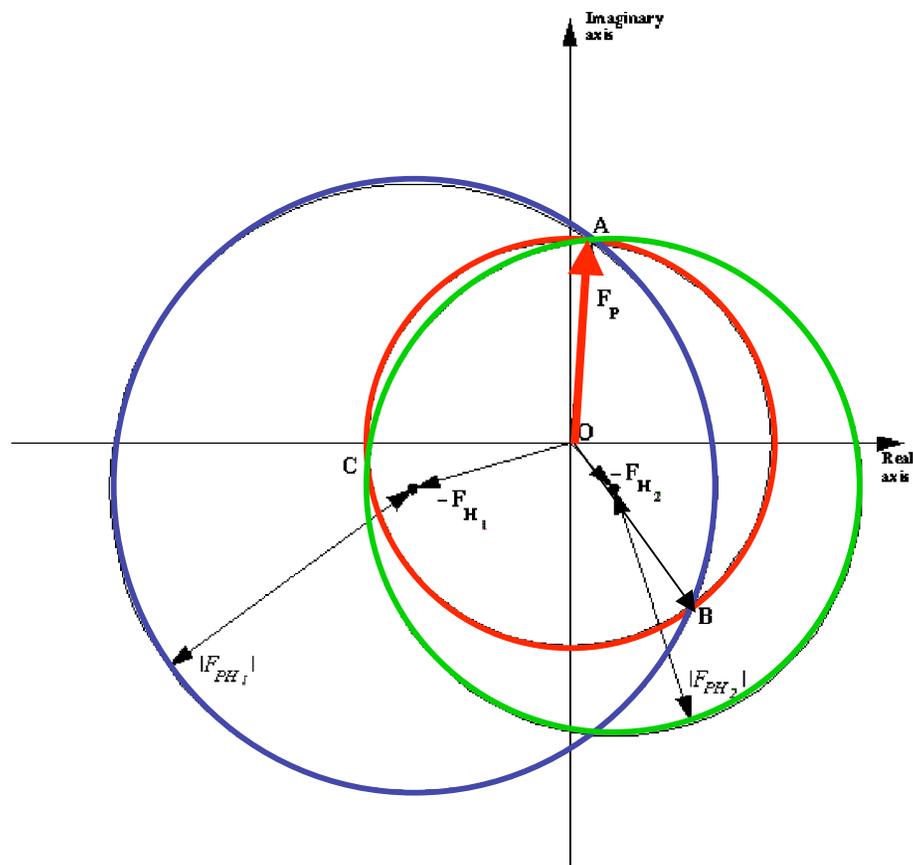
Phasing diagram for SIR



CCP4: (basic math for crystallographers)

<http://www.ccp4.ac.uk/dist/html/pxmaths/index.html>

Phasing diagram for MIR



CCP4: (basic math for crystallographers)

<http://www.ccp4.ac.uk/dist/html/pxmaths/index.html>

Phasing diagram for SIRAS

